

# Selective Dissolution of A-Site Cations in $\text{ABO}_3$ Perovskites: A New Path to High-Performance Catalysts\*\*

Wenzhe Si, Yu Wang, Yue Peng, and Junhua Li\*

**Abstract:** Selective dissolution is a common corrosion process in dealloying in which an alloy is immersed in acid to remove the active element, leaving behind an inert constituent. We introduce this technique into the treatment of oxide catalysts. A three-dimensionally ordered macroporous  $\text{LaMnO}_3$  perovskite has been prepared and treated with diluted  $\text{HNO}_3$  to selectively remove La cations, acquiring a novel  $\gamma\text{-MnO}_2$ -like material.  $\text{LaMnO}_3$  is not a satisfactory catalyst on CO oxidation. Upon the removal of La cations, the obtained sample showed a significantly higher CO oxidation catalytic activity ( $T_{50}=89^\circ\text{C}$ ) than the initial precursor  $\text{LaMnO}_3$  ( $T_{50}=237^\circ\text{C}$ ) and ordinary  $\gamma\text{-MnO}_2$  ( $T_{50}=148^\circ\text{C}$ ). A large surface area, a high degree of mesoporosity, excellent low-temperature reducibility, and especially improved surface oxygen species are deduced to be responsible for CO oxidation at lower temperatures.

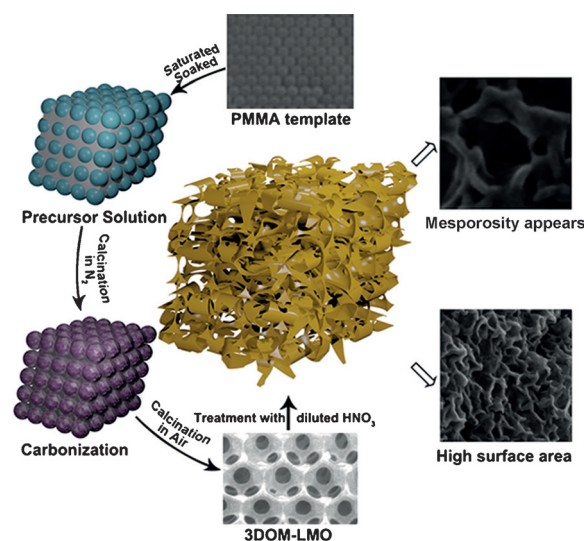
The design of high-performance catalysts by facile and universal methods has always been an important goal of research on heterogeneous catalysis.<sup>[1]</sup> A common technique in other fields, if applied in the preparation of catalysts, may have a significant effect in improving catalytic activities. Selective dissolution is such a common approach in dealloying, involving immersing an alloy in some kind of acid for a particular time to remove an active element, leaving behind an inert constituent.<sup>[2]</sup> Erlebacher and co-workers selectively dissolved Ag from Ag-Au alloys with  $\text{HNO}_3$ , resulting in the formation of a nanoporous sponge Au with a high surface area.<sup>[2a]</sup> To our best knowledge, selective dissolution has not been applied in heterogeneous catalysis until now.

Perovskites, which can be expressed with the general formula  $\text{ABO}_3$ , usually exhibit unsatisfactory performance on heterogeneous catalysis. The general explanation is that the perovskites usually possess low surface area, and the native surfaces are preferentially occupied by A-site cations, which are not catalytically active.<sup>[3]</sup> If A-site cations can be selectively removed from  $\text{ABO}_3$ , the active B-site cations

will remain, and the catalytic activity of the obtained material may increase significantly.

In the  $\text{ABO}_3$  structure, A and B cations are 12-fold coordinated and 6-fold coordinated with oxygen anions, respectively.<sup>[3a,4]</sup> A–O bonds are longer and have a higher surface energy than B–O bonds, which means that A–O bonds are easier to be attacked.<sup>[5]</sup> Herein, we investigate the possibility of selective dissolution on A-site cations from  $\text{LaMnO}_3$  (LMO) perovskites by acid treatment. A three-dimensionally ordered macroporous (3DOM) LMO material was first prepared to increase the contacting surface areas of LMO with acid. After selective dissolution, a novel manganese oxide was born with large surface area and plenty of mesoporosity. What is more, the amount and mobility of surface oxygen species were increased, which provided the novel catalyst an excellent catalytic activity on CO oxidation. Our findings suggest a new avenue for manganese oxides to improve their catalytic activities.

The treatment route for the selective dissolution is shown in Scheme 1. 3DOM-LMO was prepared by employing



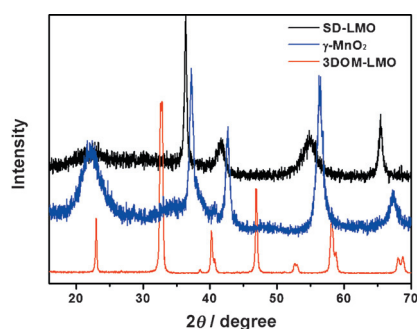
**Scheme 1.** Representation of the selective dissolution procedure.

polymethyl methacrylate (PMMA) microspheres as the hard template according to a previously reported procedure.<sup>[6]</sup> Then the 3DOM-LMO sample was immersed in dilute  $\text{HNO}_3$  until no obvious bubble was emerging. After selective dissolution, the sample was washed and dried. The obtained product was denoted as SD-LMO. A  $\gamma\text{-MnO}_2$  catalyst was prepared by hydrothermal synthesis as a contrast. The details of sample preparation and the acid treatment process are

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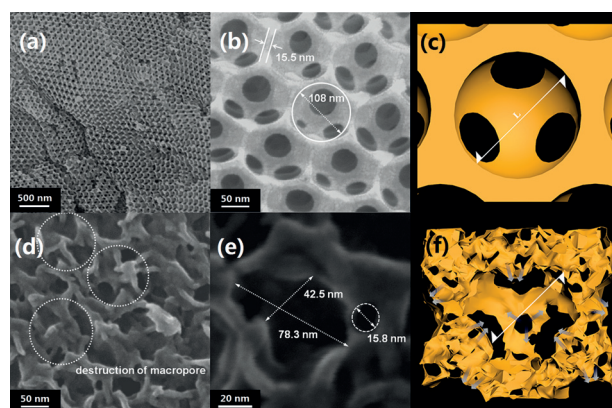
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**Figure 1.** XRD patterns of the three samples. From bottom to top: 3DOM-LMO (JCPDS No. 50-0299),  $\gamma$ -MnO<sub>2</sub> (JCPDS No. 14-0644), and SD-LMO.

given in the Supporting Information. Figure 1 shows the X-ray diffraction (XRD) results of the three samples. The XRD pattern of 3DOM-LMO shows a highly crystalline character, indexed as the perovskite phase. After the acid treatment, the diffraction peaks assigned to perovskite phase disappeared, and a  $\gamma$ -phase MnO<sub>2</sub> was born. This suggests that the dilute HNO<sub>3</sub> can selectively dissolve La cations in LMO perovskites, leaving behind MnO<sub>2</sub>. However, a small amount of La cations still existed with good distribution in the SD-LMO sample, according to ICP (Table 1) and TEM mapping results (Supporting Information, Figure S2). The residual La cations could attract O anions, and the Mn–O bonds would be weakened and become longer. Then the crystal size of the SD-LMO would be larger than that of the traditional  $\gamma$ -MnO<sub>2</sub>. As a result, a downshift of diffraction peaks for SD-LMO can be found, compared to the  $\gamma$ -MnO<sub>2</sub> in the XRD patterns.

The morphological changes of the samples could be obviously revealed in Figure 2. The 3DOM-LMO sample exhibited a high-quality 3DOM structure. Furthermore, the macropore size and wall thickness of 3DOM-LMO were  $108 \pm 10$  and  $15.5 \pm 2$  nm, respectively. After the acid treatment, destruction of the 3DOM structure could be observed (Figure 2d,e). The original smooth surface of the macropores became completely rumpled and plenty of mesoporosity appeared, which significantly increased the surface area of SD-LMO. The size of the mesoporous structure was determined to be  $15.8 \pm 5$  nm (Figure 2e). The results from the SEM images were consistent with those from the N<sub>2</sub> adsorption–desorption isotherms and pore-size distributions of these samples (Supporting Information, Figure S1). The morphological change with the acid treatment could be attributed to the shrinkage in the volume of the treated



**Figure 2.** SEM images and representations of a)–c) 3DOM-LMO and d)–f) SD-LMO.

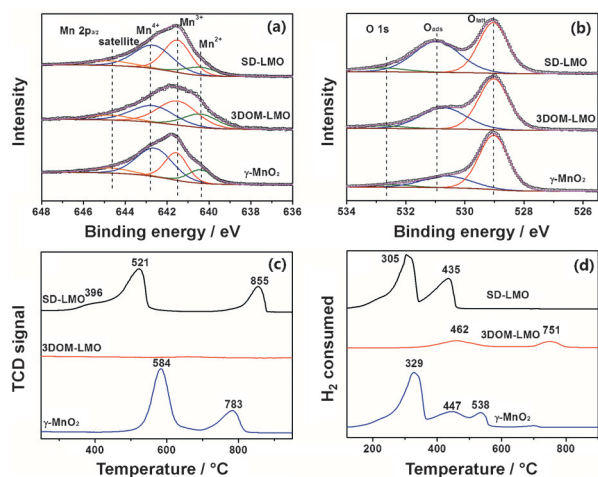
material (Figure 2c,f). Theoretically, there is 23.9 % shrinkage in volume from LaMnO<sub>3</sub> to  $\gamma$ -MnO<sub>2</sub>. Many gaps emerged as a result of the shrinkage occurring in the SD-LMO sample, which brought about many nanopores. A porous framework and high surface area are significant to improve the catalytic performance, which favor the accessibility of the reactant molecules to the catalysts.<sup>[7]</sup>

Mn 2p<sub>2/3</sub> and O 1s XPS spectra were used to confirm the changes of electronic structure after the acid treatment, and the compositions of the surface elements are summed up in Table 1. The asymmetrical XPS peaks can be decomposed by the curve-fitting approach. As shown in Figure 3a, the surface Mn<sup>4+</sup>/Mn<sup>3+</sup> molar ratio increased from 3DOM-LMO to SD-LMO after the acid treatment, which was caused by the following reaction:  $2\text{Mn}^{3+}_{\text{solid}} \rightarrow \text{Mn}^{4+}_{\text{solid}} + \text{Mn}^{2+}_{\text{liquid}}$ .<sup>[8]</sup> This reaction proceeded weakly in the presence of dilute HNO<sub>3</sub>. Moreover, the Mn<sup>4+</sup>/Mn<sup>3+</sup> molar ratio in SD-LMO (1.02) was less than that in  $\gamma$ -MnO<sub>2</sub> (1.64), which was resulted from the remaining La cations in SD-LMO. The charge imbalance caused by additional La cations in SD-LMO can be neutralized by either Mn<sup>3+</sup> or adsorbed oxygen (O<sub>ads</sub>).<sup>[9]</sup> The Mn<sup>4+</sup>/Mn<sup>3+</sup> molar ratio was reduced and the O<sub>ads</sub> was increased after the acid treatment, which could also be confirmed by O 1s XPS results (Figure 3b).

The O<sub>2</sub>-TPD experiments were carried out to further detect the changes of oxygen species after the acid treatment. The oxygen desorption at low temperature (< 400 °C) is ascribed to O<sub>ads</sub>, and the lattice oxygen (O<sub>latt</sub>) starts to be released at high temperature (> 400 °C).<sup>[10]</sup> From Figure 3c, the oxygen desorption at low temperature can only be

**Table 1:** ICP, surface element compositions, BET surface areas, H<sub>2</sub> consumption, CO oxidation activity, and apparent activation energies (*E<sub>a</sub>*) of the three samples.

	ICP		XPS		BET surface area [m <sup>2</sup> g <sup>−1</sup> ]	H <sub>2</sub> consumption [mmol g <sup>−1</sup> ]	CO oxidation activity and apparent activation energy		
	La/Mn molar ratio	Mn <sup>4+</sup> /Mn <sup>3+</sup> molar ratio	O <sub>ads</sub> /O <sub>latt</sub> molar ratio				<i>T</i> <sub>10</sub> [°C]	<i>T</i> <sub>50</sub> [°C]	<i>E<sub>a</sub></i> [kJ mol <sup>−1</sup> ]
SD-LMO	0.04	1.02	0.97		245.7	9.65	34	89	26.02
3DOM-LMO	1.18	0.75	0.63		22.4	2.04	169	237	50.47
$\gamma$ -MnO <sub>2</sub>	0	1.64	0.36		68.9	10.68	92	148	38.91

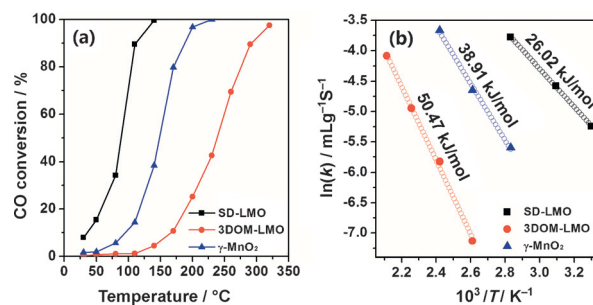


**Figure 3.** a) Mn  $2p_{3/2}$  XPS spectra, b) O  $1s$  XPS spectra, c)  $O_2$ -TPD profiles, and d)  $H_2$ -TPR profiles of the three samples.

observed in SD-LMO. The amount of  $O_{ads}$  in SD-LMO was much more than those in the other samples, which was consistent very well with the O  $1s$  XPS results. Furthermore, the  $O_{latt}$  desorption temperature of SD-LMO (521 °C) was lower than that of  $\gamma$ - $MnO_2$  (584 °C). It is indicated that the  $O_{latt}$  in SD-LMO is released in a facile manner. The reason is that the Mn–O bonds in SD-LMO were weakened and got longer as a result of the residual La cations, which caused the Mn–O bonds easier to be broken.

Figure 3d illustrates the  $H_2$ -TPR profiles of the three samples, and their quantitative analysis results are listed in Table 1. In the case of SD-LMO sample, there were two main reduction peaks at 305 and 435 °C, corresponding to a total  $H_2$  consumption of  $9.65 \text{ mmol g}^{-1}$ . According to the results reported previously, the reduction process of manganese oxide could be reasonably divided into two sections: 1)  $Mn^{4+} \rightarrow Mn^{3+}$ , and 2)  $Mn^{3+} \rightarrow Mn^{2+}$ .<sup>[10b]</sup> The total  $H_2$  consumption of SD-LMO was less than that of  $\gamma$ - $MnO_2$  ( $10.68 \text{ mmol g}^{-1}$ ). This indicated that the average valence of Mn cations in SD-LMO was lower than that in  $\gamma$ - $MnO_2$ . This was also proved by Mn  $2p_{3/2}$  XPS results. Besides, the reduction temperature of SD-LMO (305 °C) was lowest among the three samples, which suggested the best low-temperature reducibility and  $O_{latt}$  mobility.<sup>[11]</sup> This is in good accordance with  $O_2$ -TPD results.

The catalytic performances of the three samples on the oxidation of CO are shown in Figure 4. For the conversion of CO into  $CO_2$ ,  $T_{10}$  (the temperature of the conversion 10%) for samples 3DOM-LMO,  $\gamma$ - $MnO_2$ , SD-LMO are 169, 92, and 34 °C;  $T_{50}$  (the half conversion temperature) are 237, 148, and 89 °C, respectively. The catalytic activities increased in the sequence of 3DOM-LMO <  $\gamma$ - $MnO_2$  < SD-LMO. The surface areas significantly influence the catalytic activities. The SD-LMO owned the largest surface area among the three samples. Therefore the apparent activation energy ( $E_a$ ) was calculated to eliminate the effect of the surface area. The sample with lower  $E_a$  value can oxidize CO more easily.<sup>[12]</sup> From Figure 4b, the  $E_a$  value of SD-LMO was smaller than those of the others. Based on the above analysis, CO



**Figure 4.** a) Activity profiles and b) Arrhenius plots of CO oxidation over the three samples.

oxidation might proceed more readily over the SD-LMO sample. The difference in  $E_a$  is most likely related to the difference in surface oxygen species. In catalytic reaction, especially CO oxidation, the Langmuir–Hinshelwood mechanism and the Mars-van Krevelen mechanism are generally proposed to be responsible for this reaction system.<sup>[13]</sup> CO is oxidized mainly by  $O_{ads}$  in the Langmuir–Hinshelwood mechanism and by  $O_{latt}$  in the Mars-van Krevelen mechanism, respectively. Thus both  $O_{ads}$  and  $O_{latt}$  play a very important role in catalytic reaction. After the acid treatment, the residual La cations not only increased the amount of  $O_{ads}$ , but also made the  $O_{latt}$  easier to be released, improving the  $O_{latt}$  mobility on the catalyst. The two mechanisms would be enhanced at the same time. Thus the SD-LMO showed the best catalytic performance. The previously reported  $E_a$  values of the various catalysts for CO oxidation are summarized in the Supporting Information, Table S1.<sup>[14]</sup> The  $E_a$  value of the SD-LMO catalyst was much lower than most of the other manganese oxide catalysts. The results from the kinetic investigations confirm that SD-LMO expressed high catalytic performance on CO oxidation at low temperatures.

In summary, we have developed a facile method to synthesize  $\gamma$ - $MnO_2$ -like catalyst utilizing LMO perovskites as initial precursor. The novel catalyst exhibited large surface area, plenty of mesoporosity, and excellent low-temperature reducibility. In particular, the amount of  $O_{ads}$  and the mobility of  $O_{latt}$  were increased significantly after the acid treatment. All of these provided the novel catalyst with an outstanding catalytic activity on CO oxidation. Our findings not only suggest a new avenue for perovskite-based manganese oxides to improve their catalytic activities on heterogeneous catalysis, but also supply a method to synthesize novel  $\gamma$ - $MnO_2$ -like materials, which may possess huge potential in other fields.

**Keywords:** CO oxidation · heterogeneous catalysis · manganites · perovskite phases · selective dissolution

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